

It may be assumed that the elongations observed correspond to plate formation in the solid-solution matrix, in which case we estimate these regions to be about 40 Å across and four atomic layers thick.

The structural changes in the Cu-5 at.% Al crystal are shown by the way in which the intensities are redistributed: the maximum intensity in the central part of the 020 reflexion increases to 120 per unit value and that of $\bar{1}11$ decreases to 60 per unit value. The intensity change may be attributed to the formation of Al atom clusters which distort the regularity of the matrix (Cu atoms are appreciably smaller than Al atoms). This causes distortions which, in their turn, change intensities.

A further increase in the alloying-element concentration leads to a change in the diffuse intensity distribution. The shapes of iso-intensity curves for Cu alloyed with 10 and 14 at.% Al are shown in Fig. 2(d) and (e) respectively. Between the $\bar{1}11$ and 020 sites there are elongations of weak intensity. Measurement between the $\bar{1}11$ and 020 sites as a function of scattering angle (or displacement vector \mathbf{g}) shows that the intensity drops in proportion to $1/g$ (a log plot, Fig. 3), as expected for large distortions (Krivoglaz, 1967). It is therefore reasonable to expect that an increase in the alloying-element concentration would be accompanied by appreciable distortions in the solid-solution lattice.

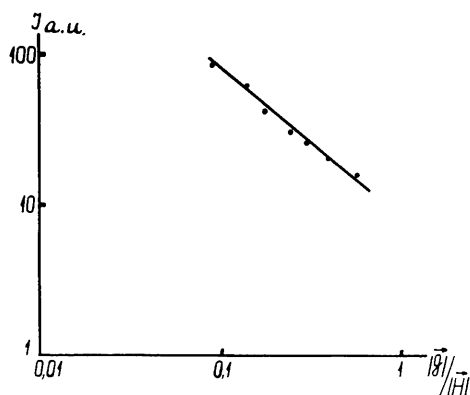


Fig. 3. Dependence of diffuse scattering intensity on displacement wavevector $|g|$.

These distortions might be associated with the formation of atom conglomerations within the solid-solution structure which lose coherence with the matrix lattice. This is supported by separation of sites $\bar{1}11$ and 020 (Fig. 2d,e).

In addition, the X-ray pictures of Cu-10 and 14 at.% Al display new diffuse reflexions which are absent at lower Al concentrations. As seen in Fig. 1, the six diffuse reflexions observed are near the primary beam at the vertices of a perfect hexagon whose sides are formed by the diffuse background in the shape of hyperbolae pointing towards the primary beam. The intensity of the six spots is roughly similar to that of the four diffuse reflexions considered earlier.

It is difficult to propose an unambiguous interpretation of the new diffuse reflexions. They may arise from the new configurations of Al atoms about Cu atoms. A configuration was considered (Epperson, Fürrohr & Ortiz, 1978) in which four Al atoms occupy a tetrahedron centred on the Cu atom with four Al atoms at the vertices. This corresponds to the trigonal symmetry of the six diffuse reflexions. It may be assumed that the new reflexions together with the $\bar{1}11$, 020 reflexions' separation indicate that the new regions appear. At growing concentrations these regions become nucleation centres for the new α_2 phase.

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Comments on *The rigid-body vibrations of molecules in crystals* by Burns, Ferrier & McMullan (1967).

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Abstract

The rigid-body analysis of the thermal vibrations in seventeen molecular structures by Burns, Ferrier & McMullan [*Acta Cryst.* (1967), **22**, 623–629] has been repeated and major discrepancies have been observed. The sources of

errors have been diagnosed and the relevant comments together with the new list of results are reported.

The criterion put forward by Burns *et al.* (1967) (BFM1) for the verification of the assumed rigid-body motion of molecules has been used by various authors in their work

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Radii of gyration and scattering curves of hollow bodies of homogeneous electron density: errata. By YUZURU HIRAGI and SHOJI IHARA, *Institute for Chemical Research, Kyoto University, Gokasho, Uji 611, Kyoto-fu, Japan*

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Abstract

Equations (15), (16), (17), (18), and (20) in the paper by Hiragi & Ihara [*Acta Cryst.* (1981), **A37**, 378–382] are incorrect. These equations should be as follows:

$$F = 4A^2 C \left[\sum_{k=1}^n f \left(s_1, \varphi - \frac{2\pi}{n} k \right) \right] \Psi_C, \quad (15)$$

$$f(s_1, \beta) = \left[s_1^2 A^2 \left(\tan^2 \frac{\pi}{n} \sin^2 \beta - \cos^2 \beta \right) \right]^{-1} \times \left\{ \exp(is_1 A \cos \beta) \times \left[i \cot \beta \sin \left(s_1 A \tan \frac{\pi}{n} \sin \beta \right) - \tan \frac{\pi}{n} \cos \left(s_1 A \tan \frac{\pi}{n} \sin \beta \right) \right] + \tan \frac{\pi}{n} \right\}. \quad (16)$$

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$$V = 2\pi ABC, \quad (17)$$

$$F = \frac{4\pi ABC J_1(s_1, K)}{s_1 K} \Psi_C, \quad (18)$$

$$F = 4\pi ABC \frac{\sin(sL) - sL \cos(sL)}{(sL)^3}. \quad (20)$$

Equations (15) and (16) in the original article lead to incorrect scattering intensity, whereas equations (17), (18), and (20) give the correct but unnormalized value. The figures in the article were calculated with the correct equations and hence need no alteration.

All information is given in the *Abstract*. The authors thank Dr P. Martel for pointing out the errors.

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Commission on Journals

Decisions taken at meetings in Ottawa, August 1981

The attention of authors planning to submit papers to *Acta Crystallographica* or *Journal of Applied Crystallography* is drawn to the following decisions taken by the Commission on Journals at meetings held in Ottawa, 14–16 August 1981. These and other revisions in editorial policy since 1978 will be published in a new version of *Notes for Authors*, which is presently in preparation.

International Symbols for Units

The Commission has recognized that, although multiples of 10^3 are the preferred prefixes in the SI System of Units, the centimetre is not prohibited by the SI system and therefore density and absorption coefficients may be given in units of g cm^{-3} and cm^{-1} respectively, if authors so wish. In all other cases, however, authors are asked to use the recommended prefixes of decimal multiples and submultiples of the SI units rather than using ' $\times 10^n$ '.

Structural Papers

Estimated standard deviations for B_{eq} . The requirement of estimated standard deviations on equivalent values of the